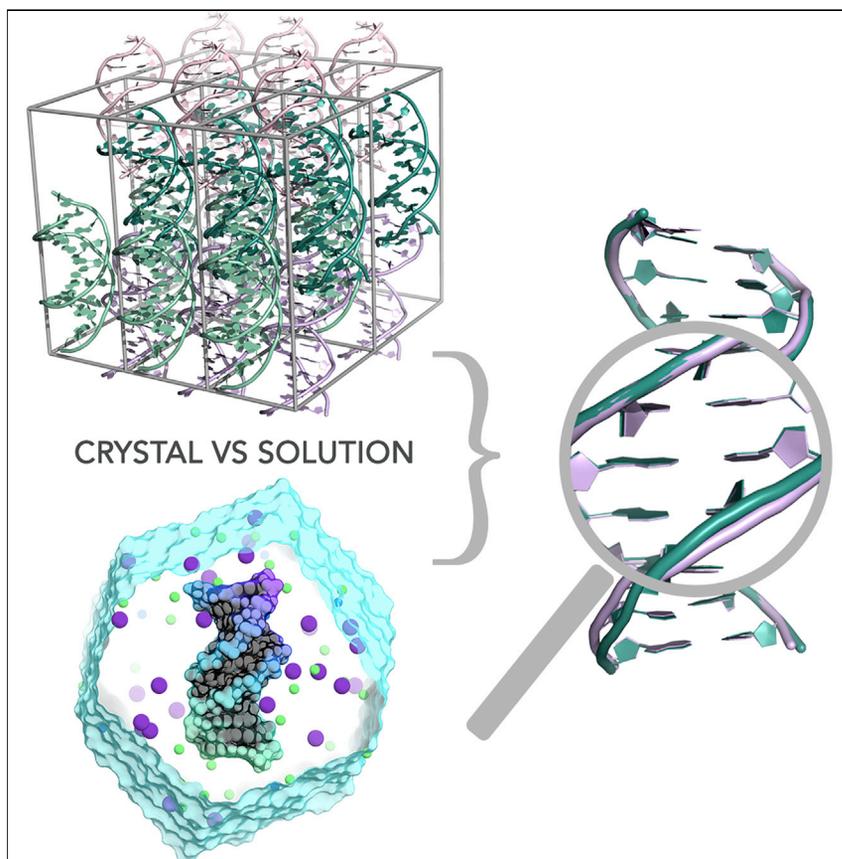


Article

An In-Depth Look at DNA Crystals through the Prism of Molecular Dynamics Simulations



Although X-ray crystallography is the most commonly used method for biomolecular structure determination, it is still unclear how stable crystals form. By using computational modeling of known crystal forms of the Drew-Dickerson dodecamer, we reveal the nature of intermolecular forces leading to stable crystals. We also explore the effects of crystallizing agents, highlighting the role of spermine in the stabilization of specific space groups. Lastly, we describe the difference in biophysical properties of DNA in solution and crystals.

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HIGHLIGHTS

Stable simulations of Drew-Dickerson dodecamer crystals (B-DNA) on the μs timescale

Analysis of all the space groups in which Drew-Dickerson dodecamer was resolved

Molecular description of the role of spermine in the stability of crystals

Detailed analysis of crystals' biophysical properties in comparison with solution

Article

An In-Depth Look at DNA Crystals through the Prism of Molecular Dynamics Simulations

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SUMMARY

X-ray crystallography is the primary tool for biomolecular structural determination. However, contacts formed through the crystal lattice are known to affect structures, especially for small and flexible molecules such as DNA oligomers, by introducing significant structural changes in comparison to solution. Furthermore, why molecules crystallize in certain symmetry groups, which role crystallization additives play, and whether they are just innocuous and unspecific crystallization catalysts remain unclear. By using one of the currently best-performing DNA force fields and applying significant computational effort, we described the nature of intermolecular forces that stabilize B-DNA crystals in various symmetry groups and solvent environments with an unprecedented level of detail. We showed a tight coupling between the lattice stability and the type of crystallization additives and that certain symmetry groups are stable only in the presence of a specific additive. Additives and crystal contacts induce small but non-negligible changes in the physical properties of DNA.

INTRODUCTION

Ever since Linus Pauling laid the foundations of structural biology, X-ray crystallography has been the cornerstone method for solving biomolecular 3D structures. Over 130,000 models derived from X-ray data and deposited in the Protein Data Bank (PDB) (out of ~147,000 models) are the best witnesses of the power of this technique, which has also become the gold standard for validating other structural methods. However, just as any other method, X-ray crystallography has its own shortcomings, and obtaining diffractable crystals is not always an easy task.^{1–3} Through robotic equipment, numerous variants of crystallization buffers are tested until a suitable one is found.⁴ However, why a given buffer promotes crystallization, what its influence is on the symmetry of the unit cell, as well as the overall structure and physical properties of the biomolecule, is typically unknown.

Thousands of X-ray-derived DNA structures have been crucial in understanding the fine details of isolated and protein-bound DNAs.^{5–7} Unfortunately, artifacts in DNA crystals can lead to conformations that are otherwise undetected in solution. Even in ideal cases, such as the Drew-Dickerson dodecamer (DDD),^{8,9} where crystal structures resemble the solution ones reasonably well, it is unclear how crystallization conditions modulate the symmetry space in which the molecule crystallizes, the fine details of the structure, and the physical properties of the DNA.^{1,2} Knowing the molecular interactions that stabilize the crystals would help to understand the actual properties of DNA in such systems and the mechanisms through which highly charged molecules are packed in severely crowded conditions, e.g., the cellular nucleus.

The Bigger Picture

Almost 90% of biomolecular structures deposited in public databases to date have been determined by X-ray crystallography. The method owes its success partly to the fact that many materials can form crystals; however, it is difficult to know beforehand which specific conditions could facilitate their formation. Thus, optimal crystallization conditions are determined by gradually changing the buffer composition, temperature, and pressure in a trial-and-error manner.

Considering how time consuming and costly such an approach is, the field would greatly benefit from knowing how external conditions determine the stability and specific symmetry of given biomolecular crystals, thereby enhancing the efficiency of the crystallization process.

Computational methods—such as molecular dynamics simulations—show great potential in this respect, and by studying the behavior of DNA in crystals and the effects crystallization additives have on crystal stability, we lay the foundation for such studies.

Here, we used a new generation of supercomputers and a recently developed DNA force field^{10–12} to analyze the nature of DDD in three known crystal lattices and to explore the effects of crystallization additives on the stability of the crystals and properties of the DNA.^{13–15} Through extensive unbiased molecular dynamics (MD) simulations on the multi-microsecond timescale, we demonstrated how the stability of DNA crystals depends on subtle interactions between the packed DNA molecules and the components of the crystallization buffer. We obtained stable atomistic simulations of DNA crystals in biologically relevant timescales^{16–21} in various symmetry groups (in which DDD has been crystallized) and in different solvent environments, which allowed us to understand with a high level of detail the nature of intermolecular interactions that stabilize the crystals. Through the analysis of numerous simulations, we characterized the physical properties of DNA in a highly crowded environment and how the presence of additives affects them.

RESULTS AND DISCUSSION

Are the Simulations of DNA Crystals Stable in the μ s Regime?

The Drew-Dickerson dodecamer is a prototypical B-DNA molecule whose structure has been determined by X-ray crystallography in various space groups and under diverse conditions^{13–15} as well as through high-quality NMR data in solution. We chose to perform our simulations on structures crystallized in three different space groups: P3₂12, P2₁2₁2₁, and H3 (PDB: 1EHV, 1FQ2, and 463D, respectively). We followed the multistep protocol developed by the Case group,¹⁷ which includes a careful stabilization of the internal pressure by calibrating the number of water molecules in the system (see the [Supplemental Experimental Procedures](#) for details; [Table S2](#)). Once the internal pressure was adjusted, we ran microsecond long simulations of each crystal, which contained 27, 24, and 36 copies of DDD ([Table 1](#)). To our surprise, two of our systems showed a systematic degradation of the crystal lattice, as shown by the center-of-mass (COM) displacements of each duplex with respect to the ideal positions in the crystal ([Figure 1](#)). The distortion of the lattice was particularly pronounced along the z axis and got worse with simulation time for 1EHV and 1FQ2, whereas the 463D system was completely stable through the trajectory. Not only was the lattice intact in 463D, but the MD conformations sampled in the helical space on the base-pair step level were in complete agreement with the average X-ray values ([Figure 2](#)), thus establishing that our force field is fully capable of representing local and global features of crystal structures.

We thus centered our efforts on stabilizing the other two crystals by starting first with the smallest system: 1EHV. We simulated a larger crystal (81 instead of 27 DDD molecules) with five different Na/KCl concentrations ([Table 1](#)), ranging from 15 to 800 mM (i.e., up to five times the assumed physiological salt concentration of the cellular nucleus), without being able to stabilize it ([Figure S1](#)). Considering that the increase in crystal's size had little effect on its stability ([Figure S2](#)), we decided to proceed with the smaller system (27 DDD copies). The addition of 400 mM MgCl₂ produced some improvement in the lattice integrity, but its degradation was still evident during the 4- μ s-long simulations ([Figure S3](#) and [Supplemental Experimental Procedures](#) for more details). We thus checked whether the global degradation of the crystals was due to internal distortions of the double helices by analyzing the 2D distributions of the base-pair-step helical parameters from MD simulations. We obtained a good agreement between the values calculated from MD simulations and the X-ray structures ([Figure S4](#)), which discards the notion that the lattice distortions were caused by the deterioration of duplex geometries. Even the flexible ends (residue pairs 1–24 and 12–13), which exhibit fast but moderate fraying in solution,¹¹

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Table 1. Studied Systems and Simulated Conditions

System Name	PDB	Environment	Space Group	No. of Unit Cells	No. of DNA Duplexes	Salt Type	[Salt] (mM)	Ratio of dsDNA to Spm	Temperature (K)	Equilibration Time (ps)	Simulation Time (μ s)
1EHV sol	1EHV	solution	–	–	1	Na/KCl	400	–	284	600	1
15 mM	1EHV	crystal	P3 ₂ 12	3 × 3 × 3	81	KCl	15	–	284	600	1
200 mM	1EHV	crystal	P3 ₂ 12	3 × 3 × 3	81	Na/KCl	200	–	284	600	1
400 mM	1EHV	crystal	P3 ₂ 12	3 × 3 × 3	81	Na/KCl	400	–	284	600	4
600 mM	1EHV	crystal	P3 ₂ 12	3 × 3 × 3	81	Na/KCl	600	–	284	600	1
800 mM	1EHV	crystal	P3 ₂ 12	3 × 3 × 3	81	Na/KCl	800	–	284	600	1
1EHV Na/KCl	1EHV	crystal	P3 ₂ 12	3 × 3 × 1	27	Na/KCl	400	–	284	600	4
1EHV Mg	1EHV	crystal	P3 ₂ 12	3 × 3 × 1	27	MgCl ₂	400	–	284	600	4
1EHV Mg hex	1EHV	crystal	P3 ₂ 12	3 × 3 × 1	27	MgCl ₂ ·6H ₂ O	400	–	284	600	4
1EHV experimental V 1:6	1EHV	crystal	P3 ₂ 12	3 × 3 × 1	27	SpmCl ₄	–	1:6	284	6,000	1
1EHV 1:9	1EHV	crystal	P3 ₂ 12	3 × 3 × 1	27	SpmCl ₄	–	1:9	284	6,000	1
1EHV 1:12	1EHV	crystal	P3 ₂ 12	3 × 3 × 1	27	SpmCl ₄	–	1:12	284	6,000	1
1FQ2 sol	1FQ2	solution	–	–	1	Na/KCl	400	–	295	600	1
1FQ2 1:6 sol	1FQ2	solution	–	–	1	SpmCl ₄	–	1:6	295	600	1
1FQ2 Na/KCl	1FQ2	crystal	P2 ₁ 2 ₁ 2 ₁	3 × 2 × 1	24	Na/KCl	400	–	295	600	4
1FQ2 Na/KCl equ	1FQ2	crystal	P2 ₁ 2 ₁ 2 ₁	3 × 2 × 1	24	Na/KCl	400	–	295	6,000	1
1FQ2 1:3	1FQ2	crystal	P2 ₁ 2 ₁ 2 ₁	3 × 2 × 1	24	SpmCl ₄	–	1:3	295	6,000	1
1FQ2 1:6	1FQ2	crystal	P2 ₁ 2 ₁ 2 ₁	3 × 2 × 1	24	SpmCl ₄	–	1:6	295	6,000	1
1FQ2 1:9	1FQ2	crystal	P2 ₁ 2 ₁ 2 ₁	3 × 2 × 1	24	SpmCl ₄	–	1:9	295	6,000	1
463D sol	463D	solution	–	–	1	Na/KCl	400	–	293	600	1
463D	463D	crystal	H3	2 × 2 × 1	36	Na/KCl	400	–	293	6,000	4

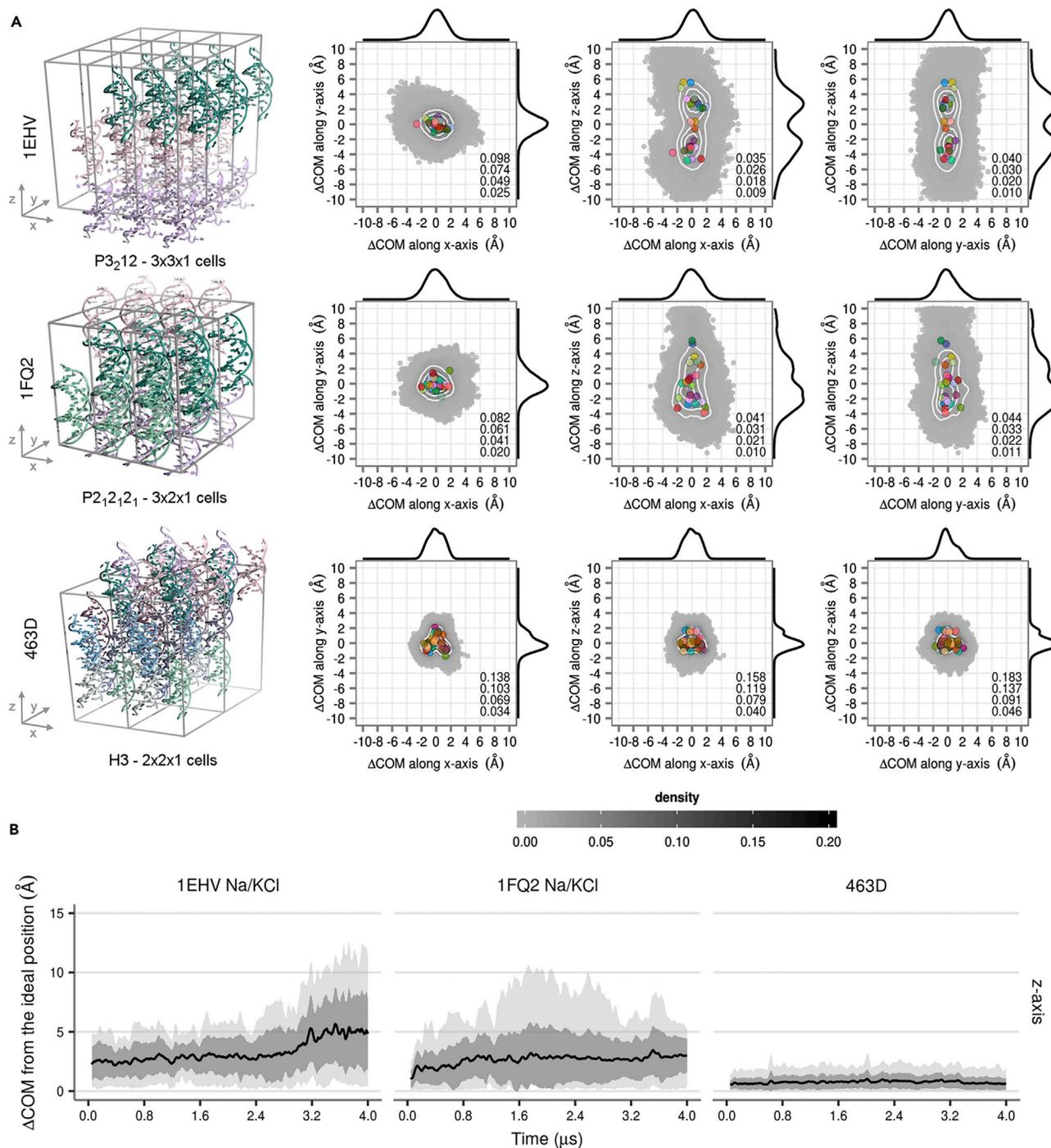


Figure 1. Stability of Three Different Supercells of Drew-Dickerson Dodecamer Crystals

(A) COM displacements from the ideal crystallographic positions are shown for each duplex in the simulated supercell systems (PBD: 1EHV, 1FQ2, and 463D) along principal axes. Points are colored in grayscale according to the smooth 2D densities, estimated by fitting the observed distributions to a bivariate normal kernel (evaluated on a square grid of 90×90 bins). Density distributions for each axis are also shown in the margins. Four isodensity curves are shown in white and are quantified on the bottom right side of each plot. The average positions of the duplexes are given as colored circles. (B) COM displacements in time along the z axis are shown for all three systems. Mean values are shown as black lines, standard deviations are shown as dark grey ribbons, and extreme values are shown with light grey ribbons (lower panels).

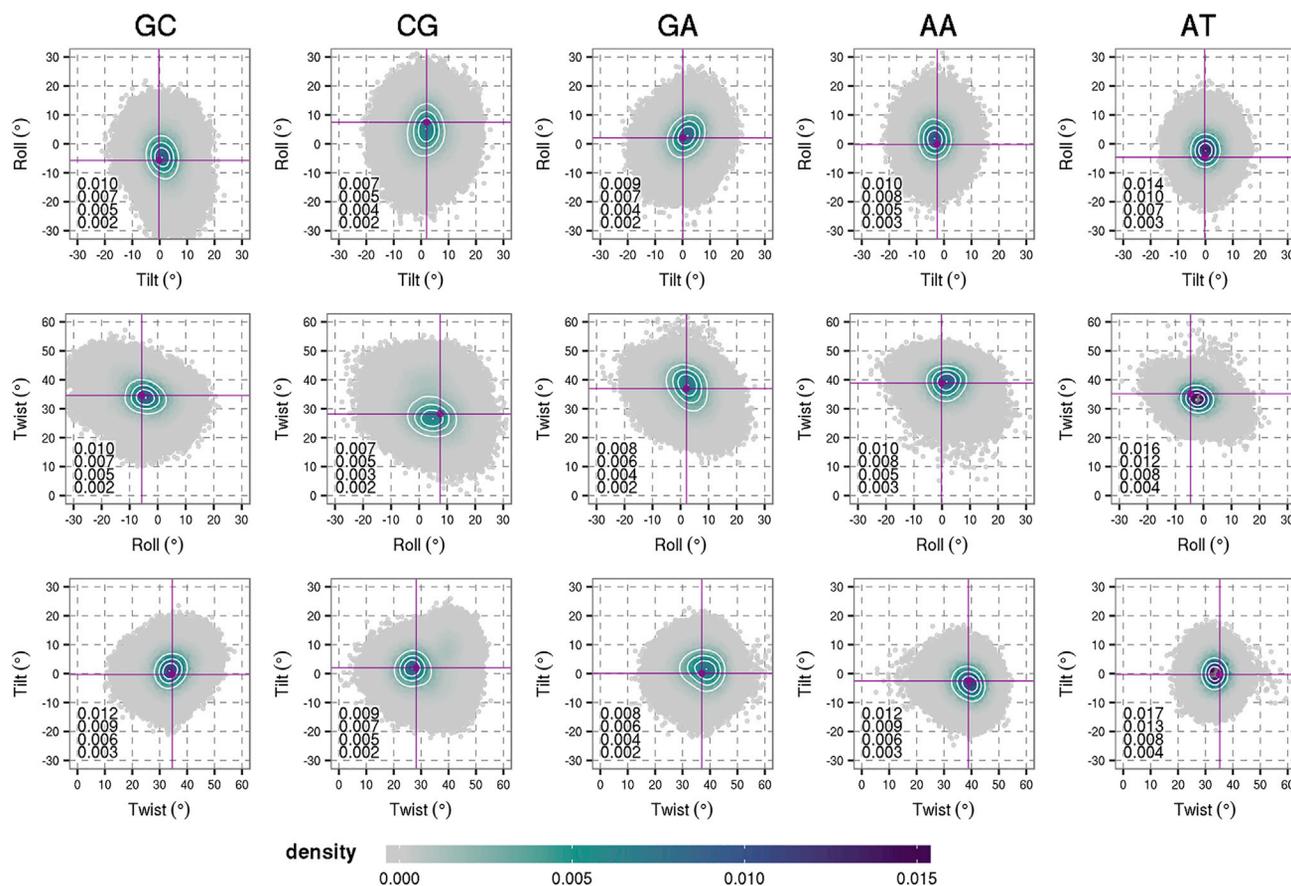


Figure 2. Rotational Helical Parameters for the 463D System

Rotational helical parameters (roll, twist, and tilt) of all the duplexes in the 463D system (given for distinct base-pair steps in DDD after removing the ends). The values calculated from the X-ray structure are shown with purple lines. Points are colored according to the smooth 2D densities, estimated by fitting the observed distributions to a bivariate normal kernel (evaluated on a square grid of 90×90 bins). Four isodensity curves are shown in white and are quantified on the bottom right side of each plot.

had in most cases an average RMSD with respect to the X-ray structures under 4 \AA , oscillating between 3D conformations compatible with the experimental electron density (Figure S5). We obtained similar results when trying to stabilize, without success, the $P2_12_1$ 1FQ2 crystal (data not shown), which allowed us to discard the possibility of a force field artifact that is space-group specific.

How Important Are the Buffer Components in the Stabilization of DNA Crystals?

Given that we had discarded any obvious explanation for the degradation of the crystal lattice, we focused our attention on the chemical additives used to obtain the DDD crystals in the studied space groups. We noticed that the 463D system was crystallized without spermine (SPM),¹⁵ whereas both 1EHV¹⁴ and 1FQ2¹³ crystals were formed in the presence of this polyamine (+4 charge at pH 7). SPM is normally found in all eukaryotic cells and has been used as an “inert molecular glue” for obtaining thousands of DNA crystals in all the major forms (A, B, Z, etc.).^{19,20,22} However, the molecular basis of its mechanism of action is mostly unknown, and in fact, SPM electron densities are often absent from the crystal. To check whether the lack of SPM in our simulations was the reason of the degradation of $P3_212$ and $P2_12_1$ simulations, we repeated the simulations for both crystals at three different

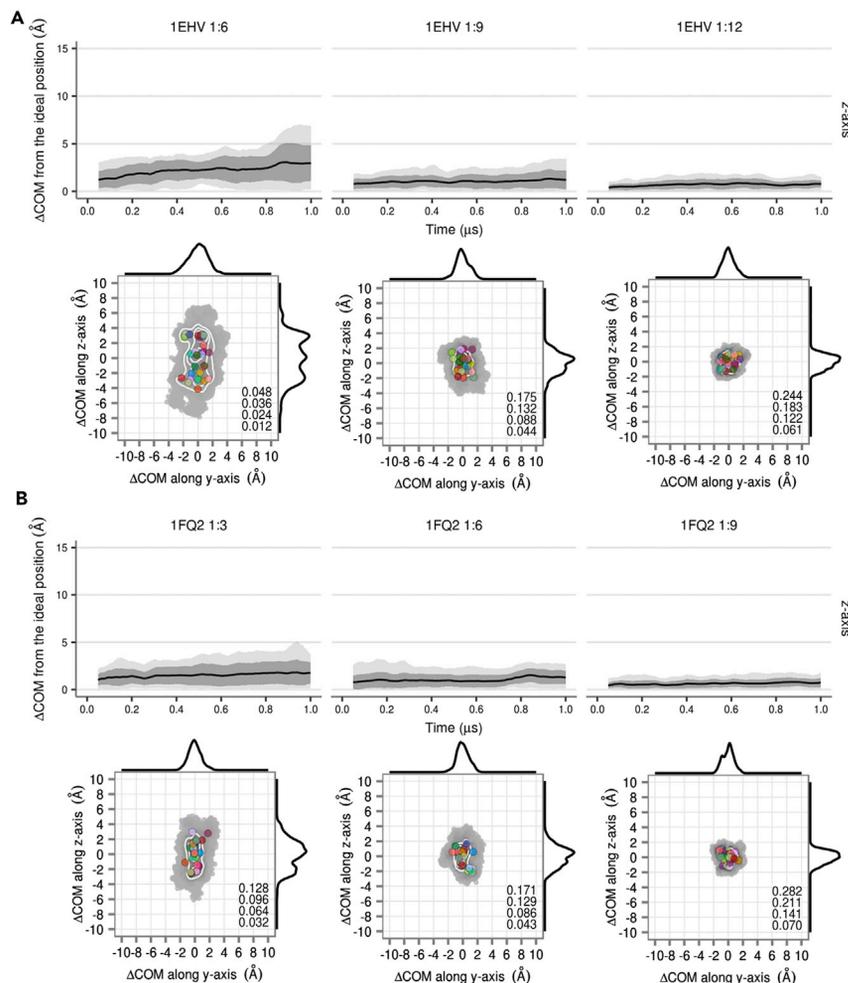


Figure 3. Effect of Spermine on the Stability of 1EHV and 1FQ2 Systems

COM displacements from the ideal crystallographic positions for different SPM concentrations in (A) 1EHV and (B) 1FQ2 systems. The upper panels show the COM displacements in time along the z axis, where mean values are shown as black lines, standard deviations are shown as dark gray ribbons, and extreme values are shown with light gray ribbons. The lower panels show COM displacements for each duplex along the y and z axes, and the average positions of the duplexes are given as colored circles. Points are colored in grayscale according to the smooth 2D densities, estimated by fitting the observed distributions to a bivariate normal kernel (evaluated on a square grid of 90×90 bins). Density distributions for each axis are also shown in the margins. Four isodensity curves are shown in white and are quantified on the bottom right side of each plot.

SPM concentrations ranging from 1:3 (three SPM molecules per duplex) up to 1:12 (Table 1). At last, the lattice integrity for both space groups was now preserved in a consistent concentration-dependent manner (Figure 3). At the medium SPM concentration tested, 1:9 for 1EHV and 1:6 for 1FQ2, the observed stability is already comparable to the one obtained for 463D with no perceptible drift of the crystal lattice along the y or z axes (Figures S6 and S7). Clearly the “inert molecular glue” has a major role in preserving the integrity of some of the crystal lattices.

Similarly to 463D, once the crystals were properly stabilized, an impressive agreement between the X-ray structures (1EHV and 1FQ2) and the MD crystals was obtained. This is visible when comparing the rotational helical space (Figure 4),

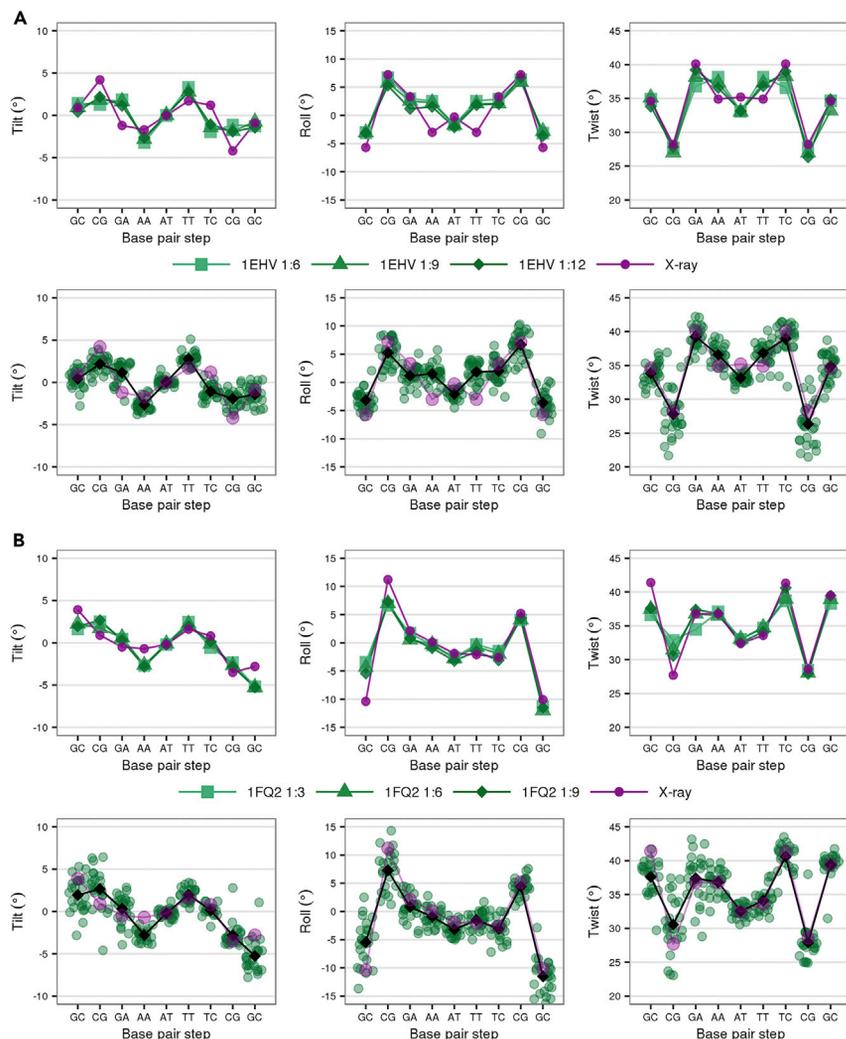


Figure 4. Effect of Spermine on Rotational Helical Parameters for 1EHV and 1FQ2 Systems

Rotational helical parameters (roll, twist, and tilt) (given for all base-pair steps in DDD after removing the ends) for different SPM concentrations in (A) 1EHV and (B) 1FQ2 systems. The upper panels show average values across all duplexes for varying SPM concentrations (colored green), and the purple line give values calculated for the X-ray structure. The lower line give the average values for each duplex in the systems with the highest SPM concentration (shown as green circles), whereas the average over all duplexes is shown as a black line and the X-ray values are in purple.

or the groove dimensions, a property that has always been difficult to reproduce accurately by modern force fields (Figure S8). The dynamics of the end residues also seem to be stable according to their RMSD values, which are typically below 2 Å with respect to the crystal position (Figures S9 and S10). In the end, to illustrate the observed structural agreement, we also compared the average structures obtained from crystal MD simulations with the deposited X-ray structure for the three systems stabilized in the μ s regime, as shown in Figure S11.

To follow each ion along the trajectories of each duplex, we used the curvilinear helicoidal coordinate (CHC)²³ method as implemented in the Canion module of Curves+, which allowed us to calculate ion populations and concentrations in the inner and outer areas of major and minor grooves of DNA duplexes (see Supplemental

Experimental Procedures).^{24,25} We found that the sequence-dependent binding sites observed for K^+ and the amino groups of SPM were the same in solution and in crystals, although the sequence dependence was weaker in crystals and the cation distributions seemed to be fuzzier (Figure S12).¹¹ The comparison of aqueous simulations with and without SPM (left column in Figure S12) showed that SPM mainly localizes outside the grooves, which was also previously observed in crystal simulations.^{19,20,22,26} Interestingly, the interiors of the grooves are depleted of cations in all crystal simulations compared to the solution phase.

We also analyzed the cation environment in all the individual duplexes present in each crystal. First, a clear similarity was found in the cation distribution in all the duplexes of a given crystal, confirming the robustness of the average results discussed here (see Figures S13–S15). Second, SPM distribution was similar around the duplexes in simulations of two different space groups (PDB: 1EHV and 1FQ2), indicating that the ion distribution is independent of the crystal packing (see Figures S14 and S15). Moreover, SPM mostly preferred the exterior of the central part of the major groove; however, upon entering the groove, SPM was located at non-central sites, which we also observed for solution simulations (see Figure S12). SPM could concentrate up to 12 M around the phosphate groups in the central portion of the duplexes, where no DNA intermolecular contacts are present. In contrast, these contacts are particularly pronounced in terminal bases (Figure S16). In summary, SPM has a tendency to move around the exterior of the duplex without any perceptible sequence dependence (as observed in NMR²⁷ and RAMAN²⁸ experiments), most likely screening intermolecular phosphate-phosphate repulsion and facilitating crystal packing. Combined with its rare long-term presence within the insides of the grooves, it is not surprising that SPM electron density is seldom captured.

Although the temperature for both experiments and MD simulations was in the 284–293 K range for 1EHV, 1FQ2, and 463D (see Supplemental Experimental Procedures), the intermolecular DNA contacts in the crystal froze the duplex structure. DNA's internal effective temperature was lowered by 200 K (with respect to the thermostat temperature) at the end of the duplexes, which are mostly rigidified, whereas we observed only a ~ 50 K decrease in the central portion of the duplex (Figure S17). Water molecules and SPM are also affected by the crowded and highly negatively charged environment produced by the packing of the backbone phosphate groups of neighboring duplexes. The rescaled diffusion coefficient²⁹ of water molecules (Figure S18 and Supplemental Experimental Procedures) was reduced by one order of magnitude—from $1.70 \cdot 10^{-5} \text{ cm}^2 \text{ s}^{-1}$ in solution DNA simulations to $1.67 \cdot 10^{-6} \text{ cm}^2 \text{ s}^{-1}$ in crystal simulations. The reduction in water mobility is visible from a 30 K decrease in its effective temperature. Furthermore, water mobility in crystals is anisotropic, with the preference for the z axis in 463D and 1EHV and the y axis in 1FQ2, following the water channels that are created in each crystal due to the orientation of the duplexes (Figure S19). The effect was even more pronounced for SPM when comparing the 1FQ2 1:6 solution simulations with the 1FQ2 crystal with the same SPM concentration: the diffusion coefficient in the crystal was reduced by 3 orders of magnitude in comparison with the solution: from $1.90 \cdot 10^{-6} \text{ cm}^2 \text{ s}^{-1}$ to $2.62 \cdot 10^{-9} \text{ cm}^2 \text{ s}^{-1}$. Although the mobility of SPM is greatly reduced in the crystal, the lack of specificity in its binding to DNA (extensive and non-specific binding outside the major groove) makes its detection in the experimental electron densities very difficult (see Figure S12). As expected, the diffusion of water and SPM in the crystal was correlated and exhibited an SPM-concentration-dependent effect, as shown in Figure S20.

Table 2. Root Mean Square Deviations (RMSD, Å) from the Deposited Crystal Structures for 463D, 1EHV, and 1FQ2

	All Heavy Atoms	Backbone	Base	All Heavy Atoms (sol)	All Heavy Atoms (sol + SPM)
463D	1.72 (0.86)	1.39 (0.99)	2.03 (0.65)	3.20 (1.20)	–
1EHV 1:12	0.96 (0.83)	1.00 (0.95)	0.90 (0.64)	3.20 (1.15)	–
1FQ2 1:9	0.57 (0.52)	0.64 (0.60)	0.47 (0.41)	1.45 (1.19)	1.49 (1.23)

The statistics in each box are heavy atoms, backbone atoms, and base atoms RMSD of average MD structure with respect to the crystal structure. The last two columns show the comparison of the average structure from the solution simulation with the deposited model. The values in parentheses exclude the terminal residues.

How Do the Structural and Dynamical Properties of DNA Crystals Compare with Solution?

We computed the global RMSD of MD simulations (crystal and solution) with respect to the experimental crystal configuration (Table 2). Our calculation expectedly showed that the average MD structure in solution deviates much more from the X-ray structure (with larger deviations in the backbone) than the average conformation from the crystal MD simulations.¹⁷ However, for all three systems, we obtained an excellent agreement for the internal 10 base pairs, with the RMSD in all cases under 0.9 Å (even 0.52 Å for 1FQ2), ratifying that MD simulations are able to capture the effect of the crystal environment on the DNA. Irrespectively of the original conformation present in the crystal, when we transferred the duplexes to solution, all the simulations converged to the same ensemble that was previously described by multi-microsecond simulations of DDD,^{11,12,30} showing a strong consensus and the lack of memory in the simulations (Figures S21 and S22). Moreover, as confirmed by WAXS experiments,²² SPM has a negligible effect on the structure of DNA in aqueous solution (Figures S21 and S22). As shown in Figure 5, the solution simulations were generally able to sample the X-ray values within the first standard deviation, indicating that the overall structures in solution and in the crystal are very similar. However, a close inspection shows small local discrepancies between the average MD solution simulations and the X-ray values at several base pairs, although these differences (Figures S23 and S24) tend to disappear when average properties of the entire duplex are computed.¹¹

We also analyzed how a crystal modifies the flexibility and dynamical behavior of the duplex. As expected, DNA in crystals was much more rigid than in solutions, which can be seen from the force constants associated with helical deformations. As expected from helical parameters, the differences are especially pronounced for the terminal residues which are much more mobile in water (Figure S25). Interestingly, the same sequence-dependent pattern of flexibility is found for both crystal and solution simulations—for all degrees of freedom and for all studied crystals. This suggests that the strong reduction of terminal movements is a consequence of the specific DNA packing required for crystallization. The analysis of the essential dynamics (see Experimental Procedures) of DNA in solution and in crystal indicates a high degree of similarity (93%–98% for solution versus solution simulations), whereas the similarity decreases to 78%–95% for crystal versus crystal MD comparisons and to 75%–84% when comparing crystal to solution (Figure S26). This finding suggests that the essential dynamics of the duplexes in crystals slightly differ from those in solution and that variability in the type of movements between different crystal lattices is larger than in solution. These results indicate that the small structural discrepancies in the average helical parameters, discussed above, arose from dynamic rather than static properties of the DNA. However, they also suggest a

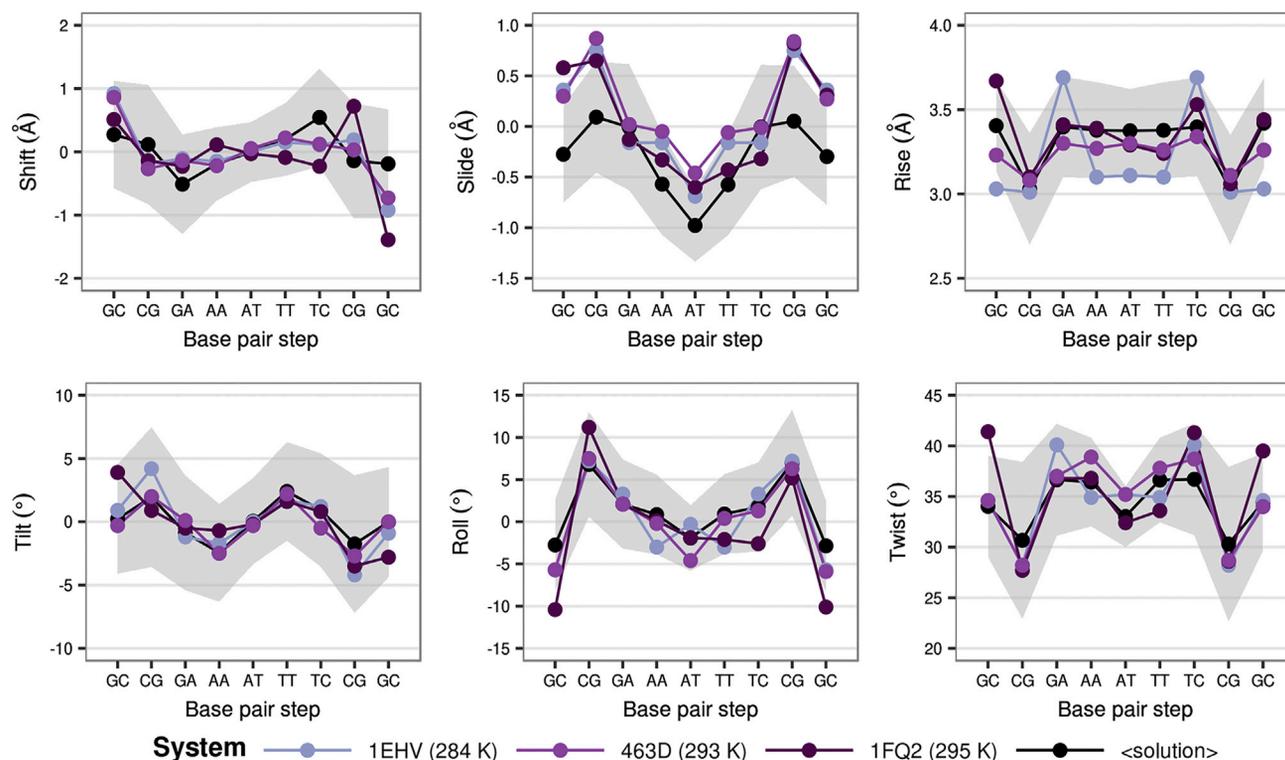


Figure 5. Comparison of Translational and Rotational Helical Parameters Obtained for Solution Simulations

Translational (shift, slide, and rise) and rotational (roll, twist, and tilt) helical parameters (given for all base-pair steps in DDD after removing the ends) for the solution simulations of single duplexes starting from different X-ray structures. The average values across the simulations are shown in black, whereas the average standard deviations are shown as gray ribbons. Values calculated from X-ray models are shown as colored lines.

reasonable maintenance of the intrinsic deformation pattern of DNA in crystals, which would explain how processes requiring deformation of the DNA can be reproduced (on longer timescales) in crystal environments, e.g., binding of small molecules in soaking experiments.^{31,32}

We finally focused our analysis on two paradigmatic structural polymorphisms that seem to be causally connected in solution, namely the backbone BI/BII polymorphism (related to ϵ and ζ torsions) and the bimodal behavior of the twist helical parameter observed for d(CpG) steps.^{11,24,33–35} BII populations were detected in both crystal and solution simulations, but the base-pair steps for which such minor populations were observed were different (Figure S27). In agreement with the experiments (X-ray and $^1\text{H}/^{31}\text{P}$ -NMR),³⁶ BII state was observed less frequently for d(CpG) and d(GpA) steps in the crystal than in the solution simulations, whereas the opposite was observed for d(GpC) steps. Quite similar results were observed for d(ApA) and d(ApT) steps. This would suggest that the differences in BII propensities between X-ray and NMR structures are not spurious effects related to the uncertainties of the refinement process but are a consequence of crystal packing. As expected from previous studies,³⁷ when transferred to the solution, the BI/BII propensities observed in the crystals converged to those in solution in the sub-microsecond timescale.

Polymorphisms at the base level (i.e., the existence of two stable conformational sub-states) were much more difficult to detect experimentally.³⁸ On the one hand, nuclear Overhauser effects often lack the required resolution to introduce dual

restraints in the simulation procedure, and on the other hand, the “substate-washing effect” produced by the averaging of structures present in the crystal annihilate minor sub-states in the final refined structures.^{38,39} Because we have individual data for each duplex in the crystal, we can analyze minor polymorphism with high accuracy. Thus, we explored the twist polymorphism at the d(CpG) step for the three simulated crystals and compared the results with the solution trajectories. We found that the bimodal behavior was mitigated in the crystals as a result of both averaging effects and a lower propensity for the bimodal twist (Figures S28–S30).^{24,35} The two and three sub-states observed in the 2D density plots of roll versus twist and twist versus tilt, respectively, were all found among the duplexes present in the crystal units, but not simultaneously in a unique duplex as in the solution simulations. This correlated perfectly with the reduced amount of cations found inside the minor groove of d(CpG) steps in the crystal simulations²⁴ and with the smaller mobility of crystal DNA that reduces the kinetics of substate inter-conversion. These results suggest that the solvent environment created around the duplexes in the crowded crystal affects the DNA dynamics and limits its conformational space in comparison with the free DNA in solution. Therefore, taking extra caution is necessary before using X-ray data to evaluate small-scale polymorphisms—which can be crucial for DNA recognition by ligands.

Conclusions

Even with recent advances in cryoEM, X-ray crystallography remains a dominant technique in structural biology and shows no signs of slowing down. However, one of its main challenges still lies in obtaining diffractable crystals. Conditions that lead to such crystals are *a priori* unknown, and the knowledge gained in the field has been mainly empirical. Typically, several types of solvent buffers are usually tried out, but given that the electron density of ions and chemical additives is rarely observed, it has been impossible to know which buffer components finally end up in the crystal and at which concentration. For example, SPM has been used for obtaining thousands of DNA crystals, yet the molecular basis of its mechanism of action was mostly unknown until now. Furthermore, atomistic MD simulations were not able to provide relevant information on the intermolecular forces that dominate within the crystal structure because of issues with obtaining stable crystal simulations on relevant timescales. In this work, we presented the first stable multi-microsecond long MD simulations of crystals of the Drew-Dickerson dodecamer in all the symmetry groups in which it has been experimentally crystallized. We achieved this by using one of the latest-generation classical force fields and by correctly modeling the solvent environment around DNA duplexes, which allowed us to understand the importance of chemical additives in the crystallization buffer. Once the crystals were stabilized, an impressive global and local structural agreement was obtained between the modeled and experimental DNA crystals. This allowed us to unravel the interactions and physical properties of water molecules, ions, and SPM at the atomic level, giving a quantifiable overview. Water molecules and SPM diffused 10 and 1,000 times more slowly, respectively, in the crowded crystal environment than in solution. Most of the SPM molecules, which could concentrate up to 12 M around DNA, were bound to phosphate groups outside of the major groove without a specific sequence dependence. Moreover, we showed that previous discrepancies in local structure—between solution simulations and X-ray structures or between solution NMR and X-ray structures—in the description of the backbone polymorphism were mainly due to packing effects that modify the solvent-solute interactions, affecting the dynamics of the DNA and limiting its conformational space in comparison with its free behavior in solution. Our work expands the limits of the field and opens the door to further systematic studies of the effects

of crystallization conditions on the obtained structures. In the near future, MD simulations of crystals could be used for anticipating and predicting the effects of a given additive, reducing the number of trial-and-error iterations usually involved in the experimental obtainment of stable DNA crystals.

EXPERIMENTAL PROCEDURES

System Setup

X-ray structures of the Drew-Dickerson dodecamer used as the starting conformations in the unbiased MD simulations are those deposited under PDB: 1EHV (1.8-Å resolution),¹⁴ 1FQ2 (1.2-Å resolution),¹³ and 463D (1.45-Å resolution).¹⁵ Both 1EHV and 1FQ2 entries contain the full DNA duplex, while the 5' cytosines are missing in 463D and were modeled in with PyMOL v1.8.2⁴⁰ by using 1EHV as the template. The dodecahedral simulation boxes for solution systems (labeled sol) were created by solvating a single DNA duplex with TIP3P water molecules,^{41,42} leaving a 10-Å solvent layer around the duplex (~20,000 atoms). K⁺, Na⁺, and Cl⁻ ions (Smith and Dang parameters)^{43,44} were added to reach neutrality and the final concentration of 400 mM (with 4:3 potassium-to-sodium ratio). In the case of 1FQ2 1:6 sol system, six SPM molecules were randomly added prior to solvation with TIP3P water and neutrality was reached with Cl⁻ ions. For the crystal systems, the simulation boxes were built following the dimensions of the unit cells reported in the PDB using PyMOL v1.8. The Na/KCl and MgCl₂ systems were completed by first adding TIP3P water and then the ions to reach neutrality and the final concentrations listed in [Tables 1](#) and [S1](#) (using Smith and Dang^{43,44} parameters for Na/KCl systems and Allnér-Nilsson-Villa⁴⁵ parameters for MgCl₂). The SpmCl₄ and MgCl₂·6H₂O crystal systems were completed by first randomly adding SPM or hexacoordinated Mg²⁺ ions (using Yoo and Aksimentiev^{46,47} parameters; quantities in [Tables 1](#) and [S1](#)), then TIP3P waters followed by monovalent counterions. System sizes ranged from 44,000 to 60,000 atoms for the smaller crystals, whereas the 1EHV systems with 81 duplexes had ~170,000 atoms.

Relaxation and Equilibration of the Systems

Energy of the simulated systems (listed in [Table 1](#)) was initially minimized through steepest-descent energy minimization. The initial velocities for the atoms were taken from Maxwell distribution at 100 K and the systems were subsequently heated to the final temperature (corresponding to the temperatures at which the crystals were grown; [Table 1](#)) in five steps of ~50 K simulated for either 100 or 1000 ps at constant volume and temperature using velocity rescale thermostat.⁴⁸ In parallel, atomic position restraints for the DNA molecules were uniformly relaxed. To achieve internal pressure close to 1 bar in crystal simulations, additional TIP3P water molecules were randomly placed within each system, which was first minimized and equilibrated, and then 50-ns production simulations were run, in a trial and error process until a suitable pressure was obtained (see [Supplemental Experimental Procedures](#) for further details).

Production Trajectories

Once simulation conditions matched those expected in the crystal or in solution, production runs (1–4 μs long) were generated using GROMACS 5.0.4 biomolecular simulation package⁴⁹ with a 2-fs integration step and coordinate output of 10 ps. Constant volume conditions were employed for the crystal simulations, whereas the constant pressure (1 bar) was used in solution simulations. All simulations were done at a constant temperature (284–295 K depending on the reported crystallization conditions). Parrinello-Rahman barostat⁵⁰ and velocity rescale thermostat were used.⁴⁸ Periodic boundary conditions and particle mesh Ewald⁵¹ were used

in conjunction with state-of-the-art simulation conditions. DNA was described with the recently developed parmbc1 force field.¹⁰ See [Supplemental Experimental Procedures](#) for additional details of the simulations.

Analysis

Center of mass (COM) displacements were calculated after the optimal roto-translational fitting of the whole system to the ideal crystal. RMSD values were calculated for each duplex after the optimal roto-translational fitting to the original X-ray structures (see [Supplemental Experimental Procedures](#) for details). Diffusion coefficients were calculated from a linear fit of mean-square displacements over time using Einstein's equation. The effective temperature of water molecules in crystals was estimated from the corresponding diffusion coefficients which were first scaled by 1/2.56 (to account for the overestimation of the water diffusion by the TIP3P water model) as described in our previous work,²⁹ whereas effective DNA temperature was inferred at the base pair step level from the analysis of helical covariance matrices, which provided the stiffness matrices associated to harmonic deformations in the helical space.⁵² Essential dynamics was determined by the diagonalization of the Cartesian covariance matrices. The similarities in essential deformation modes were computed with Hess⁵³ and Pérez et al. metrics⁵² without considering the capping base pairs. Additional details on the analysis and the associated packages are shown in [Supplemental Experimental Procedures](#).

DATA AND SOFTWARE AVAILABILITY

All trajectories reported here are deposited in the BigNASim⁵⁴ database and can be freely downloaded: <http://www.irbbarcelona.org/BIGNASim/>.

SUPPLEMENTAL INFORMATION

Supplemental Information includes 30 figures, 2 tables, and Supplemental Experimental Procedures and can be found with this article online at <https://doi.org/10.1016/j.chempr.2018.12.007>.

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AUTHOR CONTRIBUTIONS

Original Idea, P.D.D.; Conceptualization, A.K., P.D.D., and M.O.; Investigation, A.K.; Formal Analysis, A.K. and P.D.D.; Visualization, A.K.; Resources, P.D.D. and M.O.; Writing—Original Draft, P.D.D.; Writing—Reviewing and Editing, A.K., P.D.D., and M.O.; Supervision, P.D.D. and M.O.; Funding Acquisition, M.O.

DECLARATION OF INTERESTS

The authors declare no competing interests.

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